

## Poly(vinyl alcohol)-hyaluronic Acid Membranes for Wound Dressing Applications: Synthesis and *in vitro* Bio-Evaluations

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Physically crosslinked poly(vinyl alcohol)-hyaluronic acid (PVA-HA) hydrogel membranes composed of different amounts of HA were prepared by freeze-thawing (F-T) method. F-T cycle was repeated for three consecutive cycles. HA was chosen and routinely utilized in the local treatment of chronic wounds, because of its advantages as, HA is endogenous and biodegradable polymer. Physicochemical properties of PVA-HA membranes such as, gel fraction (GF), swelling, mechanical properties, hydrolytic degradation and *in vitro* bio-evaluation tests were investigated. Results revealed that introducing HA into PVA structure affected significantly the physicochemical properties of membranes than the pristine PVA, because of its crosslinking interaction with PVA. With the increase of HA content in PVA hydrogel membranes, GF and mechanical stability of PVA-HA membranes decreased. However, the swelling behavior, mechanical flexibility, protein adsorption and hydrolytic degradation of PVA membrane increased. The HA content < 20% in PVA hydrogels showed high cell viability (%) and no toxicity was observed using microculture tetrazolium assay (MTT-assay). However, less cell viability was determined with high HA incorporation. PVA-HA-ampicillin free showed antimicrobial activity against *Candida albicans* as a result of HA presence. Thus, ampicillin-loaded wound dressing with PVA-HA membranes could be used as promising materials with easy forming and biologically evaluated for wound care.

**Keywords:** poly(vinyl alcohol), hyaluronic acid, hydrogel membranes, freezing-thawing, wound dressing

### Introduction

Hydrogel membranes have been employed previously as essential materials for fabricating wound dressing materials, which were invented in 1989 by Rosiak *et al.*<sup>1</sup> However, some of these membrane materials showed a drawback like low mechanical properties resulting in sticking to the wound surface or damaged under stress or stretching, which did not satisfy as perfect dressing requirements.<sup>2</sup> Therefore, the ideal wound dressing materials should meet the following conditions: (i) maintain a local moist environment, (ii) keep the wound protected

from any side contamination, (iii) good surface absorbance for wound fluids, (iv) reduce the wound surface necrosis, (v) avoid the wound dryness, (vi) stimulate the growth factors, and also be (vii) elastic, non-antigenic and biocompatible/biodegradable material.<sup>3-5</sup> Accordingly, wound dressings are utilized predominantly to improve the various wound healing stages and innovate the proper healing environments. Thus, the wound surface must be covered by dressing materials to enhance and grow the healing process. The most important behind the wound care is to promote for a quick wound healing accompanied with acceptable cosmetic appearance. Furthermore, wound healing is regarded as a special biological step connected to the skin growth and regeneration processes.<sup>4</sup>

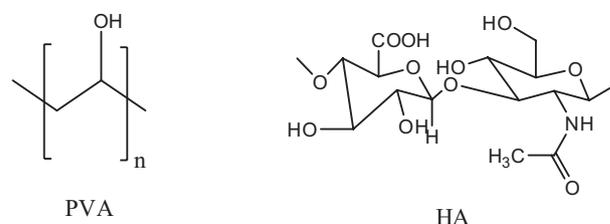
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Poly(vinyl alcohol) (PVA) as a hydrophilic polymer is water soluble (Figure 1).<sup>2</sup> PVA hydrogels have been previously utilized intensively for several biological applications, due to its biological advantages such as: nontoxic, non-carcinogenic, biodegradable and bio-adhesive characteristics with the ease of processing.<sup>2,6,7</sup> As a result of the latter features, PVA is able to simulate natural tissues and easily to be accepted in the body implantation. PVA gels have been applied in different biomedical application sites, such as contact lenses, the lining for artificial hearts, wound dressing, and drug delivery. Generally, PVA polymer can crystallize upon cooling from the melt, as known crystallization process to form ordered region called lamellae networks.

Although the precise mechanism of gelation in physical PVA gels is still not well understood,<sup>6</sup> previous experiments have used the semi-crystallization process using annealing amorphous PVA film. Thus, the basic source of the entangled PVA stability was the crystalline structure formed during annealing process,<sup>8-10</sup> and freeze-thawing (F-T) repeated cycles method based on the crystallization of PVA molecules.<sup>11,12</sup> In recent decades, the need of physical crosslinked gels has been potentially increased. The reason that behind avoiding the use of chemical crosslinker is the fact that these chemicals are not only predominantly toxic compounds, which can be removed or somewhat extracted from final gels before application, but also can affect the biological integrity of the substances when entrapped (e.g., proteins, drugs, and cells). Accordingly, the physical crosslinking method in particular F-T cycles method based on crystallization, has been preferred for PVA crosslinking as free from solvent or crosslinkers comparable with the common chemical crosslinking method.<sup>6,13</sup>

Hyaluronic acid (HA) is a high molecular weight biopolysaccharides, it is a natural linear dipolysaccharides consists of  $\beta$ -(1,4)-linked *D*-glucuronic acid and  $\beta$ -(1,3) *N*-acetyl-*D*-glucosamine units, (Figure 1). This polyanionic polymer has unique physicochemical properties and distinctive biological functions.<sup>14</sup> HA is presented in the human body specifically in neural and epithelial tissues. Thus, HA was chosen as a good blended polymer with PVA, due to its biological, endogenic and natural origin. Some recent biomedical applications of HA included ophthalmic surgery, arthritis treatment, polymeric scaffolds for wound healing, tissue engineering, cartilage repair, and drug delivery,<sup>15</sup> and it has been used also as components for implant or scaffold materials.<sup>16-18</sup> Previously, natural polymers (e.g., alginate, dextran, chitosan, glucan, starch, hydroxyethyl starch, gelatin and their derivatives), and synthetic polymers (e.g., polyethylene glycol, polyvinylpyrrolidone, and poly *N*-isopropylacrylamide),

were used as blended polymers with PVA for wound dressing applications and they were recently reviewed by Kamoun *et al.*<sup>6</sup> Recently, PVA-HA derived hydrogels were recently synthesized using thiol-yne click reaction.<sup>19</sup> Also, PVA-HA microgels for biomedical applications, were previously synthesized using click chemistry method.<sup>20</sup> Herein, the present work is designed to prepare a novel blended hydrogel membranes based on PVA and varied portions of HA contents using F-T consecutive cycles for crosslinking. The designed PVA-HA membranes were tested in terms of their physicochemical and biological properties (e.g., bovine serum albumin (BSA) protein adsorption, cell viability (%), and antimicrobial activity) to be compatible as wound dressing materials in biomedical application.



**Figure 1.** Molecular structures of PVA and HA.

## Experimental

### Materials and methods

PVA (typically average  $M_w = 72,000 \text{ g mol}^{-1}$ ; 98.9% hydrolyzed) was obtained from Biochemica, Germany. HA was purchased from Shanghai Jiaoyuan industry Co., Ltd, China. Ascorbic acid ( $M_w = 176.13 \text{ g mol}^{-1}$ ; 99%), phosphate buffer saline (PBS), pH 7.4 and ampicillin sodium salt were obtained from Sigma-Aldrich Chemie GmbH, Steinheim, Germany. Distilled water was used throughout this research.

### Preparation of PVA-HA hydrogel membranes

PVA-HA hydrogel membranes were prepared by F-T cycle according to the reported slightly modified procedure of Peppas and Stauffer.<sup>21</sup> Briefly, aqueous solution containing 5% (m/v) PVA, 1% (m/v) of HA and 0.3% (m/v) of ascorbic acid as a plasticizer was carefully dissolved in distilled water. Different portions of HA contents (0%, 10%, 20%, 30%, 40%, and 50%, m/v), were mixed with calculated amounts of ascorbic acid. The aforementioned polymers solution was mixed thoroughly using ultrasonic water-bath (ultrasonic cleaner water-bath was obtained from Thermoline Scientific, Australia) at 40 °C, for 1 h

and then vortexed for two minutes to ensure the mixture homogeneity. Proper amount of this mixture (ca. 15 mL) is poured in plastic Petri dishes, followed by freezing at  $-20\text{ }^{\circ}\text{C}$  for 18 h and then thawing for 6 h at  $25\text{ }^{\circ}\text{C}$  and eventually the formed membrane was kept unfrozen for 1 h. The F-T cycles were repeated for three consecutive cycles for providing mechanically acceptable membranes are proper for further experiments. The membrane thickness was adjusted by casting 15 mL of polymeric mixture solution. The thickness of the wet membranes was between  $200\text{ }\mu\text{m}$  and  $300\text{ }\mu\text{m}$ . The obtained PVA-HA membranes were freeze dried under vacuum at room temperature to obtain the final PVA-HA membranes. The thickness of resultant dried membranes was adjusted in the range of  $150\text{-}200\text{ }\mu\text{m}$ . The resultant PVA-HA membranes are stored in plastic and air-vacuumed pages until use and tests.

## Characterizations

### Gel fraction

The obtained PVA-HA hydrogel membranes were vacuum-dried at room temperature to avoid the membrane surface shrinkage, for 6 h and weighted ( $W_0$ ), then soaked in distilled water for 24 h up to an equilibrium swelling weight for removing any leachable soluble HA parts from the membrane. The gel membrane was then dried again and weighted again ( $W_e$ ). The gel fraction (GF%) was carried out according to the method reported by Yang *et al.*,<sup>22</sup> and calculated using the equation 1.

$$\text{Gel fraction (GF\%)} = (W_e / W_0) \times 100 \quad (1)$$

where, ( $W_0$ ) and ( $W_e$ ) are the weights of hydrogel samples dried for 6 h at  $50\text{ }^{\circ}\text{C}$  before and after soaking, respectively.

### Swelling or water uptake

The swelling degree of PVA-HA hydrogel membranes was carried out in distilled water at  $37\text{ }^{\circ}\text{C}$ . The membrane samples were cut into  $2\text{ cm} \times 2\text{ cm}$  pieces and vacuum-dried at room temperature for 6 h, the weight of dried sample was determined ( $W_d$ ). The dried samples were soaked in distilled water, and incubated at  $37\text{ }^{\circ}\text{C}$ , then weighted again ( $W_s$ ) at specific interval times. The degree of swelling or water uptake of PVA-HA hydrogel membranes could be described as degree of water absorptivity of the hydrogel membrane (equation 2).<sup>22</sup>

$$\text{Water uptake (\%)} = (W_s - W_d / W_d) \times 100 \quad (2)$$

where ( $W_s$ ) is the weight of swelled sample and ( $W_d$ ) is the weight of dried sample after soaking.

### Mechanical strength

The mechanical properties of PVA-HA blend hydrogel membranes (maximum tensile strength and the elongation-at-break) have been conducted using a universal tensile test machine (Universal Testing Machine, model: AG-I /50 N-10 KN, Japan). PVA-HA membranes were cut into specific a dog-bone like-shape demission ( $5\text{ cm}$  long,  $1.5\text{ cm}$  wide at the ends and  $1\text{ cm}$  in the middle). The analysis was performed at stretching rate  $10\text{ mm min}^{-1}$ . The thickness of membrane samples were measured using an electronic digital micrometer before examination.<sup>23</sup>

### Hydrolytic degradation

The weight loss (%) against the immersion time in PBS solution was determined by the hydrolytic degradation method.<sup>24</sup> This method is based on gravimetric determination study of the weight loss (%) of the gel. Dried membrane samples with a dimension ( $2\text{ cm} \times 2\text{ cm}$ ) were weighed and immersed in  $10\text{ mL PBS}$  ( $0.1\text{ mol L}^{-1}$ , pH 7.4, at  $37\text{ }^{\circ}\text{C}$ ). The samples were removed at time intervals, and gently wiped with soft paper to remove surfaced water, and gently dried at ambient temperature, then weighed again.

### Protein adsorption study

The amount of adsorbed BSA was detected by UV-Vis spectrophotometer at  $630\text{ nm}$ , with supporting a standard calibration curve of BSA ranging from  $3.1\text{-}60\text{ mg mL}^{-1}$ . Beer's law was used to determine the exact adsorbed BSA at the membrane surface as follows below in equation 3.

$$A = a c L \quad (3)$$

where, A is the absorbance, c is the concentration, a is a proportionality constant and L is the path length which is constant.<sup>25</sup> Pieces of PVA-HA hydrogel membranes cut into ( $1\text{ cm} \times 1\text{ cm}$ ) then were immersed in  $10\text{ mL PBS}$  (pH 7.4), and incubated at  $37\text{ }^{\circ}\text{C}$  for 24 h until reach to equilibrium swelling weight. The swollen hydrogel pieces were transferred to buffer solution containing BSA ( $30\text{ mg mL}^{-1}$ ) and shacked for 4 h at  $37\text{ }^{\circ}\text{C}$ , and then the hydrogel pieces were gently removed. The protein adsorption was calculated by the difference between protein concentrations before and after immersing hydrogel pieces in protein/phosphate buffer solution using albumin reagent kit (at  $630\text{ nm}$ ), this procedure has been adapted and slightly modified from the reported procedure of Lin *et al.*<sup>26</sup>

### *In vitro* biocompatibility and cell viability testing

Human hepatoma (HepG2) and Hela cells have been chosen to investigate the cytotoxicity or cell viability (%) of PVA-HA membranes, using microculture tatrazolium assay

(MTT-assay). The cells were cultured and grown in complete RPMI 1640 media supplemented with 10% fetal bovine serum and complete Dulbecco's Modified Eagle Medium (DMEM) media supplemented with 10% fetal bovine serum for HepG2 and Hela cells, respectively, (at 37 °C in a 5% CO<sub>2</sub>, 95% humidity atmosphere). Briefly, about 10<sup>4</sup> HepG2 and Hela cells in 200 µL complete media were plated in 96-well micro-titer plates and PVA-HA membranes were added then cultured for 4 days at 37 °C. Another method, the cells were seeded in 96-well plates at a density of 10<sup>4</sup> cells *per* well and incubated overnight at 37 °C and 5% CO<sub>2</sub>, then the medium was refreshed with new media containing PVA-HA membranes. The cells were incubated for 4 days at 37 °C. After incubation, the cells were washed twice with PBS or fresh media, and 200 µL of MTT solution (0.5 mg mL<sup>-1</sup> in PBS) was added to each well. After incubation for 6 h at 37 °C, 5% CO<sub>2</sub>, the media was left-aside and the wells were dried. Formazan crystals were re-suspended in 200 µL dimethyl sulfoxide (DMSO), and then shaken for 5 min to fully dissolve formazan in the solvent. The optical density was monitored at 570 nm with a reference at 630 nm. Parallel medium containing test substances was treated the same way in the absence of cells to exclude staining effects with adding substances itself. The relative cell viability (%) compared to control wells containing cells without adding other additives was determined, as given in equation 4.<sup>27-29</sup>

$$\text{The relative cell viability (\%)} = [A_{\text{test}} / A_{\text{control}}] \times 100 \quad (4)$$

where, A<sub>test</sub> is number of cells after incubation with membranes and A<sub>control</sub> is number of initial cells before incubation with membrane samples. The dynamic fluid viscosity of culture medium DMEM was measured before and after cell culture using dynamic fluid viscometer (model: SVM 3000-Stabinger viscometer, Anton Paar, USA).

#### Antimicrobial activity test

##### Disc diffusion method

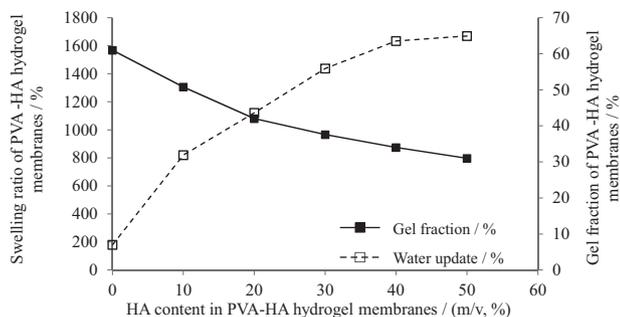
The effect of PVA-HA hydrogel disc membranes (different HA contents, e.g., 0, 10, 20, 30, 40, and 50%, m/m) against human pathogenic bacteria and fungi was examined using the universal agar diffusion method as following. Luria broth (LB) agar medium (Oxoid, England) was prepared according to the manufacture instructions and was sterilized at 1.5 psia and 115 °C for 20 min. Before solidification; the prepared sterilized medium was poured into 9 cm sterile Petri plates and kept to solidify at room temperature. Overnight cultures (18 h) of LB containing single colony of the tested pathogenic

bacteria and fungi tests as follows: Gram-negative bacteria, (e.g., *Escherichia coli* (*E. coli*), *Klebsiella pneumonia*, and *Enterobacter sp.*), Gram-positive bacteria, e.g., *Staphylococcus aureus* and fungi, e.g., *Candida albicans*, respectively were grown at 30 °C and 200 rpm, then were spread over the LB plates using sterile cotton swabs. Hydrogel disc membranes (0.7 cm diameter) were distributed on the plate's surfaces under sterile conditions using sterile forceps. For hydrogel disc samples which containing ampicillin as an antibiotic model; 0.5 mg mL<sup>-1</sup> of ampicillin sodium salt was mixed to PVA-HA blended solution, vortexed for 30 min and poured in Petri-dish for conducting F-T cycles before evaluating the antimicrobial test. The plates were then preserved at 4 °C for 1 h and then transferred to 30 °C incubator for 24 h. Finally, the formed clear zones or microbial inhibition zones were measured in centimeters and recorded precisely. This test has been conducted triplicate to calculate the inhibition zone average.

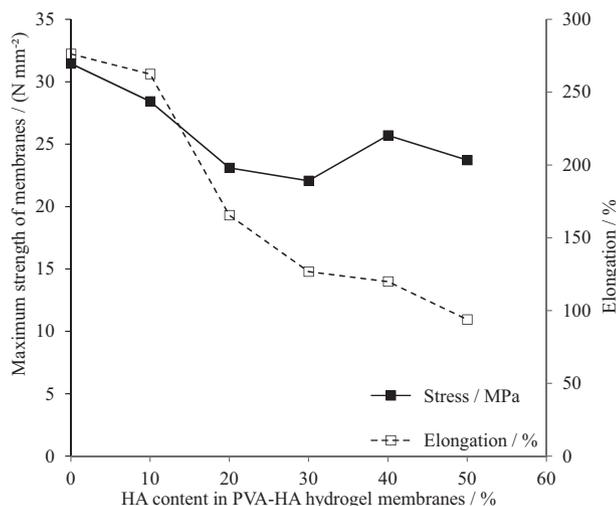
## Results and Discussion

### Swelling and gel fraction

Figure 2 shows the water uptake and the gel fraction of PVA-HA hydrogels membrane versus different HA contents using equations 1 and 2. In the light of presented swelling results, the results indicate clearly that, a significant water uptake was noticed by addition of HA (0-20 m/v, %). Moreover, a convergent increase in the water uptake was found with incorporation HA between 30% and 50%. It was found that also the maximum swelling ability of PVA-HA hydrogels increases with an increase of HA contents in PVA hydrogels. These results were attributed to the high hydrophilicity degree of HA that increases the swelling ability of obtained PVA-HA hydrogels as HA is added ceaselessly. These results are coincided with previous reported results in literatures.<sup>6,13,30,31</sup> They found that, the swelling of PVA-hydroxyethyl starch (PVA-HES) and PVA-alginate blended hydrogel membranes increased with increasing the blended polymer, e.g., HES or alginate contents, respectively. On the other side, the gel fraction of the PVA-HA hydrogel membrane decreases progressively with the increase of HA content. Generally, when the gel fraction decrease, the hydrogel strength becomes weaker, however the hydrogel flexibility is increased.<sup>32,33</sup> Therefore, HA could be used to adjust the strength and flexibility of hydrogel, because it reduces the crosslinking reaction opportunities and consequently the gel formation reduces too. These results are fully consistent with results of Zhai *et al.*,<sup>34</sup> for PVA-starch blended hydrogels and



**Figure 2.** Effect of HA contents in PVA hydrogel membranes on water uptake (%) and gel fraction (%).



**Figure 3.** Mechanical properties of PVA hydrogel membrane with respect to different ratio of HA contents.

Zhao *et al.*,<sup>35</sup> for PVA-carboxymethylated chitosan blend hydrogels.

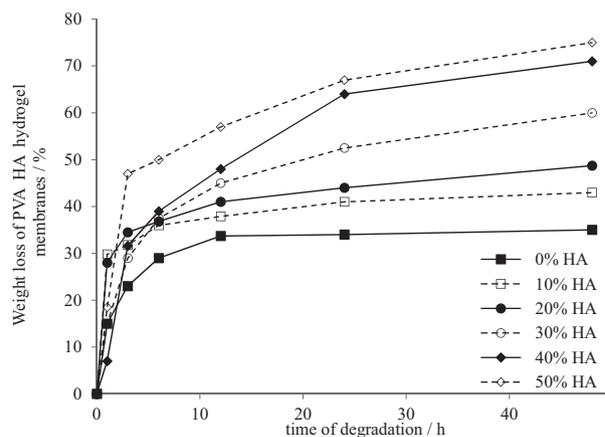
### Mechanical properties

The influence of HA content on the mechanical properties of PVA hydrogel membranes has been estimated in terms of, their tensile strength and elongation-at-break, as presented in Figure 3. The maximum tensile strength and elongation-at-break of PVA-HA hydrogel membranes, decreases clearly with increasing the HA content in membranes. As HA was blended with PVA, the crosslinking density of the gel was decreased and the mechanical flexibility increases. These results are coincided with that of Razzak *et al.*<sup>36</sup> They reported that the maximum tensile strength of PVA hydrogel deteriorated with addition of the blended polymer, owing to decreasing the crosslinking degree and density. Hwang *et al.*<sup>37</sup> demonstrated that, the maximum tensile strength of PVA hydrogel has decreased sharply with increasing dextran portions in the hydrogel. Similarly, Kenawy *et al.*<sup>13</sup> reported that the maximum tensile strength and elongation-at-break of PVA-HES

hydrogel membranes, sharply decreased with increasing HES contents. All these reported contributions refer to the addition of blend polymers into PVA might hinder the entanglement reaction, unlike the mechanical flexibility improved.

### Hydrolytic degradation

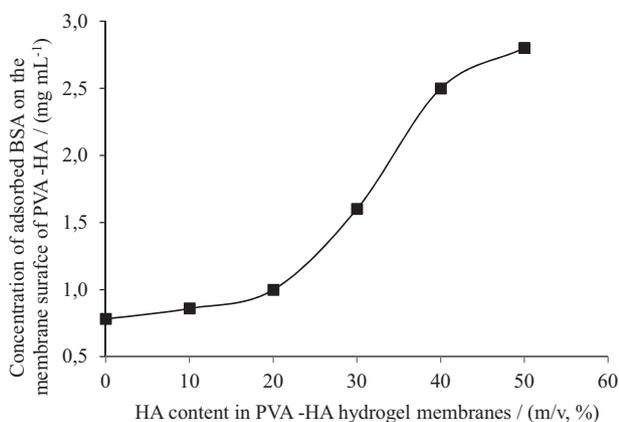
Gravimetric calculation was performed to study the hydrolytic degradation or the mass loss (%) of PVA-HA hydrogel membranes. The hydrolytic degradation of PVA hydrogel membrane as a function of different hyaluronic acid contents in membranes was conducted in PBS, as shown in Figure 4. The results exhibited that, the constant rate of hydrolytic degradation of PVA hydrogel membrane increases progressively with increasing the HA content in PVA hydrogel membranes, due to increasing of the hydrophilicity of HA parts in membrane composition. Weight loss (%) results of PVA-HA hydrogels as presented in Figure 4 are fully coincided with the swelling results as presented in Figure 2. This phenomenon can be ascribed to the degradation of PVA-HA hydrogel membranes that are mostly caused by the breaking of crosslinking segments between PVA and HA molecular structures which produce low molecular weights and degraded polymers. The weight loss results refer to that, the degradation behavior follows the second order kinetic. This also is owing to the fact that, the degradation of PVA is much slow (ca. 30% within 2 days, Figure 4) as well as previously discussed by Takasu *et al.*,<sup>38</sup> whereas the degradation of PVA-HA is much higher than the pristine PVA hydrogel (between ca. 40% and 78% within 2 days, Figure 4). In addition, as PVA and HA are nontoxic, this implies that, obtained byproducts of degraded PVA-HA moieties might be hypothesized to be nontoxic too.



**Figure 4.** Weight loss of the PVA-HA hydrogel membranes versus different degrading times in PBS (0.1 mol L<sup>-1</sup>, pH 7.4, at 37 °C), with presence of different ratios of HA in PVA membranes.

## Protein adsorption

The membrane blood compatibility is evaluated by the amount of plasma protein adsorbed onto the membrane surface. The protein adsorption onto PVA-HA blend hydrogel membranes has been conducted via *in vitro* experiment. Figure 5 shows the protein adsorption onto surface of PVA-HA hydrogel membrane as function of different HA contents in distilled water. The mechanism of protein adsorption on surface of PVA-HA membranes is owing to various types of interaction forces between protein molecules and the membrane surface, such as weak bonding (e.g., Van der Waal interactions, ionic bonding, hydrogen bonding or hydrophobic interactions) or strong chemical bonding due to chemically surface modified membranes.<sup>13</sup> Thus, in our study the clearest values of protein adsorption on PVA-HA surface have been detected with the highest values of hydrophilic surface interaction due to addition of HA portions. The results appeared that the adsorbed protein onto surface of PVA hydrogel membrane increases with increasing HA contents in PVA hydrogel membranes, and the highest values of protein adsorption on PVA-HA surface have been detected with the highest values of hydrophilic surface interaction, due to the addition of HA contents. Interestingly, the protein adsorption onto membrane surface has improved within 200% to 400%, after inclusion the HA with different portions (Figure 5). For further evidences, when the tested membrane is direct contacting with the blood, the protein is adsorbed onto membrane surface resulting in platelet adhesion and activation.<sup>39,40</sup> Since, the albumin adsorption on the synthetic surfaces could inhibit platelet activation, which does not promote clot formation. Therefore, high protein adsorption property of wound dressing materials is of the most required characteristics.<sup>41</sup> Thus, HA gave less adhesion of platelets onto artificial surfaces and high protein

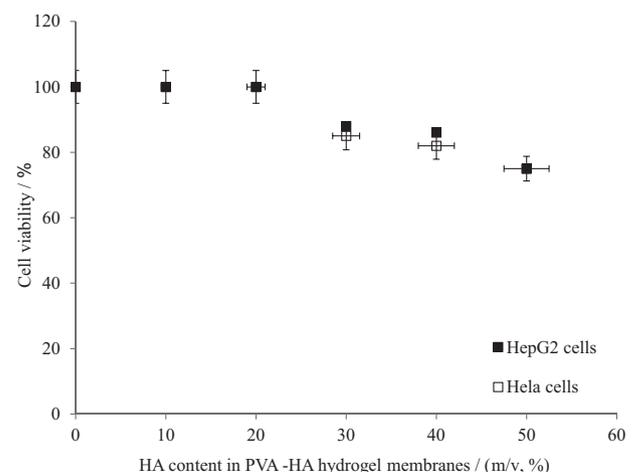


**Figure 5.** Effect of HA content on protein adsorption onto the surface of PVA-HA hydrogel membranes.

or plasma adsorption property. These results are compatible with reported results by Kim *et al.*,<sup>33</sup> and Hwang *et al.*<sup>37</sup> They proved that, the adsorption of protein increased with increasing blended alginate and dextran polymers in PVA hydrogel membranes, respectively.

## *In vitro* biocompatibility test

*In vitro* biocompatibility is principle property of any material being implanted in the biological body. The biocompatibility and cell viability (%) toward the components of PVA-HA membranes were tested *in vitro* by MTT-assay using HepG2 and Hela human cell lines (Figure 6). As shown, PVA-HA hydrogels with low HA contents (i.e., 0-20%) showed ca. 100 (%) of cell viability and non-toxicity behavior was observed. In the contrast, the cell viability (%) decreased dramatically with increasing HA contents (> 30%) in PVA membranes composition, and relatively less dead cells were observed. These results might be attributed to the released or degraded HA molecules in the culture media assuming an expanded random coil structure in physiological solution which occupy a very long domain and HA forms viscous compounds in DMEM culture media. This high viscous media might inhibit and reduce the migration, movement and cell viability (%). Thus, the fluidic viscosity of DMEM culture media increases as (0.00081, 0.0021, 0.0086, 0.0145, 0.0162, and 0.0187 Pa.s) for (0, 10, 20, 30, 40, and 50 m/v, %) of HA content), respectively. Furthermore, this allows an exceptionally high swelling ability for PVA-HA hydrogels with high HA contents, (Figure 2). These results are fully consistent with results of Schramm *et al.*<sup>42</sup>, and Becker *et al.*<sup>43</sup> They demonstrated that, the high HA content in DMEM culture media increased the viscosity, and consequently the biocompatibility of the culture media was reduced as well.



**Figure 6.** *In vitro* cells viability assay against different hyaluronic acid contents in PVA hydrogel membranes.

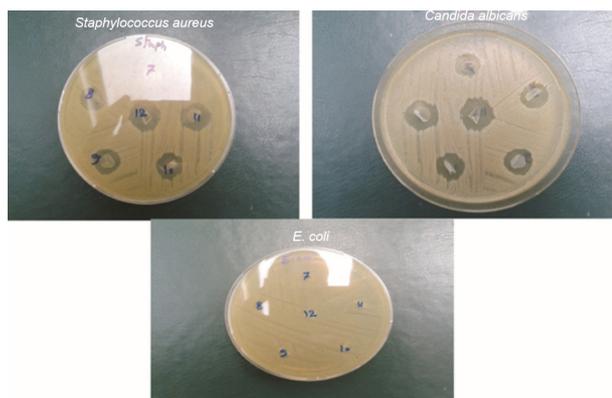
**Table 1.** Effect of various HA contents in PVA membranes composition on their antimicrobial activity

HA / %	Antimicrobial activity of PVA-HA disc membranes with ampicillin-loaded; inhibition zone diameter / cm		Antimicrobial activity of PVA-HA disc membrane without ampicillin-loaded; inhibition zone diameter / cm				
	<i>E. coli</i>	<i>Staphylococcus aureus</i>	<i>Staphylococcus aureus</i>	<i>E. coli</i>	<i>Klebsiella pneumoniae</i>	<i>Enterobacter sp.</i>	<i>Candida albicans</i>
	Gram (-)	Gram (+)	Gram (+)	Gram (-)	Gram (-)	Gram (-)	fungi
0 <sup>a</sup>	0	1.0	1.0	0	1.0	1.0	1.0
0	0	1.1	0.7 (±)	0.7 (±)	1.4 (+)	0	1.4 (+)
10	0.8	1.3	0.7 (±)	0.7 (±)	0	0	1.4 (±)
20	0	1.3	0.7 (±)	0.7 (±)	0	0	1.5
30	0	1.4	0	0	0	0	1.4
40	0	1.5	0	0	0	0	1.3
50	0	1.8	0	0	0	0	1.0

<sup>a</sup>Control assay is conducted in case PVA-ampicillin-loaded with zero-HA content in all samples.

### Antimicrobial activity

The antimicrobial activity test is a crucial parameter for wound dressing. The antimicrobial assay of PVA-HA membranes with/without ampicillin-loaded against *E. coli*, *Staphylococcus aureus*, *Klebsiella pneumoniae*, *Enterobacter sp.* and *Candida albicans* was checked. The results of bacterial inhibition zones were listed in Table 1, and their selected photographs were presented in Figure 7. An antibiotic like ampicillin or another antimicrobial drug is urgent loaded onto the dressing material to stop the microbes growth or killing them outright.<sup>44</sup>



**Figure 7.** Representative photographs of antimicrobial activities of PVA-HA membranes appearance of microbial inhibition zones formed against seeded *Staphylococcus aureus*, *Candida albicans* and *E. coli*.

Thus, most dressing materials are derived from polymeric materials loaded-antimicrobial drug, even in case the used polymeric materials have antimicrobial activity themselves, like chitosan-dressing materials which have a dual antimicrobial actions.<sup>22,45</sup> Interestingly, the results in case *E. coli*, no inhibition zone was formed even with

PVA-HA-ampicillin loaded membranes. This implies that, *E. coli* has a hard resistant against membrane compositions containing ampicillin. This was clearly evident when the control assay (PVA-ampicillin discs) exhibited also negative results against the microbial growth. Conversely, the control assay sample and PVA-HA-ampicillin-loaded could effectively constrain the growth of *Staphylococcus aureus*, and the formed inhibition zones were obviously dispersed with HA incorporation as compared with the control assay. However, these inhibition zones do not form in the absence of ampicillin-loaded membranes, (Table 1 and Figure 7). These results might be attributed to *Staphylococcus aureus* growth inhibition is owing to only the presence of ampicillin antimicrobial inhibitor and the inhibition zones were further improved with HA introduction, which is entirely opposite behavior of *E. coli*. It is exciting to find that, PVA-HA membranes without ampicillin showed a big resistance for microbial growth against *Candida albicans* and the inhibition zones were improved with HA incorporation to PVA hydrogel as found with *Staphylococcus aureus*. This result is regarded as important and remarkable results, because of *Candida albicans* is well-known the most frequently microbial can infect the body skin wound, which matches the study goal. Overall, the microbial resistance was improved for PVA-HA with certain HA content in membrane composition (up to <20%), and it was further improved with the membrane ampicillin-loaded. Unexpectedly, in case of *Klebsiella pneumoniae* and *Enterobacter sp.*, PVA-HA membranes in the absence of ampicillin do not show antimicrobial activity and no inhibition zones formed, whereas the inhibition zones were found with the control assay (Table 1). The current antimicrobial results are consistent with our published results of Kamoun *et al.*<sup>31</sup> They showed antimicrobial

activities for PVA-alginate membranes ampicillin-loaded against Gram-negative and Gram-positive bacteria after altering the alginate content in PVA membranes composition. Our results conflict slightly with the results of Abd El-Mohdy *et al.*<sup>46</sup> They revealed that the various PVP contents in PVA hydrogels showed antimicrobial activities against *E. coli*, *Bacillus subtilis* and *Staphylococcus aureus* owing to PVP has antimicrobial activity itself, and no activity against *Pseudomonas aeruginosa* was observed. Similarly, the antibacterial behavior of hydrogels based on PVA/water-soluble-chitosan was ascribed to their capability to bind the negatively charged bacteria to the positively charged amino groups of polymer in hydrogel.<sup>22</sup>

## Conclusions

In conclusion, the PVA-HA hydrogels have been successfully prepared by F-T as a physical crosslinking method. The physicochemical properties of obtained PVA-HA hydrogels have been assessed as function of different HA contents. The results showed that incorporation of HA in the F-T crosslinked PVA network appreciably affected its physicochemical properties. The swelling behavior, mechanical flexibility, hydrolytic degradation rate and protein adsorption of PVA-HA hydrogel membranes increased with increasing the HA content in hydrogel composition. However, as the gel fraction decreased, the mechanical strength of the gel was weakened but the flexibility was increased. Addition HA into PVA hydrogel membranes improved protein adsorption property onto PVA membrane surface as compared to pristine PVA membranes. Therefore, the HA content in hydrogel composition could be used to control the mechanical strength and flexibility of PVA-HA hydrogels because it reduced the crosslinking reaction and, consequently the gelation process. PVA-HA hydrogel membranes supernatant exhibited nontoxic behavior and higher cell viability with low HA contents (up to <20%), compared with the high HA content hydrogels. Interestingly, PVA-HA without-ampicillin membranes showed effective antimicrobial activity, particularly against *Candida albicans*, as a result of the HA presence. However, the same membranes showed similar antibacterial activity against *Staphylococcus aureus* especially after loading ampicillin. The PVA-HA membranes offered no microbial resistance against *E. coli*, membranes with/without ampicillin-loaded. Thus, the addition of HA to PVA hydrogels changed and improved the physicochemical properties and biological activity of membranes, compared with the pristine PVA hydrogel for wound dressing applications. Thus, it is a potential wound dressing with easy forming and improved bio-evaluations for wound dressing application.

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## References

1. Rosiak, J. M.; Rucinska-Rybus, A.; Pekala, W.; *US Patent 4,871,490* **1989**.
2. Yoshii, F.; Zhanshan, Y.; Isobe, K.; Shiozaki, K.; Makunchi, K.; *Radiat. Phys. Chem.* **1999**, *55*, 133.
3. Kannon, G. A.; Garrett, A. B.; *Dermatol Surg* **1995**, *21*, 583.
4. Lin, S. Y.; Chen, K. S.; Run-Chu, L.; *Biomaterials* **2001**, *22*, 2999.
5. Purna, S. K.; Babu, M.; *Burns* **2000**, *26*, 54.
6. Kamoun, E. A.; Chen, X.; Mohy Eldin, M. S.; Kenawy, E. S.; *Arab. J. Chem.* **2015**, *8*, 1.
7. Long, Z.; Hiroshi, M.; Maolin, Z.; Fumio, Y.; Naotsugu, N.; Tamikazu, K.; *Carbohydr. Polym.* **2013**, *53*, 439.
8. Mallapragada, S. K.; Peppas, N. A.; *Polym. Mater. Sci. Eng. Proceed.* **1995**, *73*, 22.
9. Mallapragada, S. K.; Peppas, N. A.; *J. Polym. Sci. Part B Polym. Phys.* **1996**, *34*, 1339.
10. Mallapragada, S. K.; Peppas, N. A.; *AIChE J.* **1997**, *43*, 870.
11. Hassan, C. M.; Peppas, N. A.; *Macromolecules* **2000**, *33*, 2472.
12. Papanca, A.; Valente, A. J. M.; Patachia, S.; Miguel, M. G.; Lindman, B.; *Langmuir* **2008**, *24*, 273.
13. Kenawy, E.-R.; Kamoun, E. A.; Mohy Eldin, M. S.; El-Meligy, M. A.; *Arab. J. Chem.* **2014**, *7*, 372.
14. Laurent, T. C.; Laurent, U. B. G.; Fraser, J. R. E.; *Ann. Rheum. Dis.* **1995**, *54*, 429.
15. Saetone, M. F.; Monti, D.; Torracca, M. T.; Chetoni, P.; *J. Ocul. Pharmacol. Ther.* **1994**, *10*, 83.
16. Candy, T.; Sharma, C. P.; *Biomater. Artif. Cells Artif. Organs* **1990**, *18*, 1.
17. Hong, S. R.; Lee, S. J.; Shim, J. W.; Choi, Y. S.; Lee, Y. M.; Song, K. W.; Park, M. H.; Nam, Y. S.; Lee, S. I.; *Biomaterials* **1993**, *14*, 2777.
18. Okamoto, Y.; Minami, S.; Matuhashi, A. In *Advances in Chitin and Chitosan Animals*; Brine, C. J.; Sandford, C. J.; Zikakis, J. P., eds.; Elsevier: New York, 1992.

19. Li, Y.; Wang, L.; Wu, J.; Ma, Y.; Wang, J.; Wang, Y.; Luo, Y.; *Mater. Letters* **2014**, *134*, 9.
20. Kupal, S. G.; Cerroni, B.; Ghugare, S. V.; Chiessi, E.; Paradossi, G.; *Biomacromolecules* **2012**, *13*, 3592.
21. Peppas, N. A.; Stauffer, S. R.; *J. Controlled Release* **1991**, *16*, 305.
22. Yang, X.; Liu, Q.; Chen, X.; Yu, F.; Zhu, Z.; *Carbohydr. Polym.* **2008**, *73*, 401.
23. Alencar, D. Q. H.; Humberto, G. F.; Gustavo, A. A.; Maria, M. F.; Antonio, L. B.; Julio, S. R.; *J. Biomed. Mater. Res. Part A* **2003**, *64*, 147.
24. Xiao, C.; Zhou, G.; *Polym. Degrad. Stab.* **2003**, *81*, 297.
25. Queiroz, A. C.; Santos, J. D.; Monteiro, F. J.; Gibson, I. R.; Knowles, J. C.; *Biomaterials* **2001**, *22*, 1393.
26. Lin, W. C.; Yu, D. G.; Yang, M. C.; *Colloids Surf., B* **2006**, *47*, 43.
27. El-Fakharany, E. M.; Haroun, B. M.; Ng, T. B.; Redwan, E. R.; *Protein Pept. Lett.* **2010**, *17*, 1031.
28. El-Fakharany, E. M.; Sánchez, L.; Redwan, N. A.; Redwan, E. M.; *J. Virol.* **2013**, *10*, 199.
29. Mosmann, T.; *J. Immunol. Methods* **1983**, *65*, 55.
30. Fahmy, A.; Abu-Saied, M. A.; Kamoun, E. A.; Khalil, H. F.; Youssef, M. E.; Attia, A. M.; Esmail, F. A.; *J. Adv. Chem.* **2015**, *11*, 3426.
31. Kamoun, E. A.; Kenawy, E. S.; Tamer, M. T.; El-Meligy, M. A.; Mohy Eldin, M. S.; *Arab. J. Chem.* **2015**, *8*, 38.
32. Ajjji, Z.; Othman, I.; Rosiak, J. M.; *Nucl. Instr. Meth. Phys. Res. Sect. B* **2005**, *229*, 375.
33. Kim, J. O.; Park, J. K.; Kim, J. H.; Jin, S. G.; Yong, C. S.; Li, D. X.; *Int. J. Pharm.* **2008**, *359*, 79.
34. Zhai, M.; Yoshii, F.; Kume, T.; Hashim, K.; *Carbohydr. Polym.* **2002**, *50*, 295.
35. Zhao, L.; Mitomo, H.; Zhai, M.; Yoshii, F.; Nagasawa, N.; Kume, T.; *Carbohydr. Polym.* **2003**, *53*, 439.
36. Razzak, M. T.; Darwis, D.; Zainuddin, S.; *Radiat. Phys. Chem.* **2001**, *62*, 107.
37. Hwang, M. R.; Kim, J. O.; Lee, J. H.; Kim, Y.; Kim, J. H.; Chang, S. W.; Jin, S. G.; Kim, J.-A.; Lyoo, W. S.; Han, S. S.; Ku, S. K.; Young, C. S.; Choi, H. G.; *AAPS Pharm. Sci. Technol.* **2010**, *11*, 1092.
38. Takasu, A.; Itou, H.; Takada, M.; Inai, Y.; Hirabayashi, T.; *Polymer* **2002**, *43*, 227.
39. Burkatovskaya, M.; Tegos, G. P.; Swietlik, E.; Demidova, T. N.; Castano, A.; Hamblin, M. R.; *Biomaterials* **2006**, *27*, 4157.
40. Coleman, D. L.; Gregonis, D. E.; Andrade, J. D.; *J. Biomed. Mater. Res.* **1982**, *16*, 381.
41. Dion, I.; Baquey, C.; Havlik, P.; Monties, J. R.; *Int. J. Artif. Organs* **1993**, *6*, 545.
42. Schramm, C.; Spitzer, M. S.; Henke-Fable, S.; Steinmetz, G.; Januschwski, K.; Heidusehka, P.; Geis-Gerstorfer, J.; Biedermann, T.; Bartz-Schmidt, K. U.; Szurmann, P.; *Invest. Ophthalm. Vis. Sci.* **2012**, *53*, 613.
43. Becker, L. C.; Bergfeld, W. F.; Belsito, D. V.; Klaassen, C. D.; Marks Junior, J. G.; Shank, R. C.; Slaga, T. J.; Synder, P. W.; Andersen, F. A.; *Inter. J. Toxicol.* **2009**, *28*, 5.
44. Chhatri, A.; Bajpai, A. K.; Shandhu, S. S.; Jain, N.; Biswas, J.; *Carbohydr. Polym.* **2011**, *83*, 876.
45. Kanatt, S. R.; Rao, M. S.; Chawla, S. P.; Sharma, A.; *Food Hydrocolloids* **2012**, *29*, 290.
46. Abd El-Mohdy, H. L.; Ghanem, S.; *J. Polym. Res.* **2009**, *16*, 1.

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